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## Introduction

- The solar wind (~96% H<sup>+</sup>, ~4% He<sup>2+</sup>) can impinge directly on the dayside of the lunar surface, releasing material via **ion sputtering**.
- Among other processes, this is a source mechanism in the formation of the lunar exosphere, noteworthy for the supra-thermal ejecta energies.
- Often, exosphere models rely on popular sputtering codes for input data on the sputtering contribution without further validation.
- We aim to constrain three physical quantities describing the sputtering process: angular distribution, sputtering yield and energy distribution.

## Sputtering yields

- Investigations carried out for both He and H (upper and lower figure, respectively) at solar wind energies (1 keV/nucleon)
- Influence of surface morphology: The increase in roughness when going from the **thin film** sample to the **pellet** decreases the sputtering yield and suppresses the dependence on the incidence angle
- This reduction is observed both in experiments and simulations [7]
- Red line:** simulations for porous structures modelled after [8], representative for the conditions in the **Lunar regolith**
- SRIM (dashed): previous assumptions

Sputtering yields have been severely overestimated!

## Methods

### Quartz Crystal Microbalance (QCM) [1]

- $\frac{\Delta m}{m} = -\frac{\Delta f}{f}$  [2]
- Resolves mass changes of **thin films** in real time → direct sputter yield determination
- Used sample: thin film grown from Apollo 16 #68501

### Catcher QCM [3]

- Second QCM facing the samples, collecting ejecta
- Probes angular distribution of released particles
- With calibration through the **thin film** → obtain sputter yields of **bulk samples** indirectly
- Used sample: pressed pellet [4] from Apollo 16 #68501

### Simulations – SDTrimSP [5, 6]

- Track simulated ions and their generated recoils through a solid
- Binary Collision Approximation (BCA): assumes a series of independent collisions between two nuclei
- Most basic version assumes flat, amorphous targets
- Cannot usually correctly calculate sputtering yields *ab initio* → experimental benchmarks important, ongoing debate on how to sensibly choose the variety of input parameters
- SDTrimSP-3D: three-dimensional variant allowing to untangle surface topography effects from intrinsic material properties

## Ejecta energy distributions

- Proposed method for measurements: Time of Flight – Secondary Neutrals Mass Spectrometry (**ToF-SNMS**) [9]
- Most sputtered particles neutral → ionise using a focused laser
- Timing between **ion pulse**, **ionisation** and extraction and knowledge of distance between **sample** and ionisation volume gives kinetic energy of ejecta
- Time of flight difference between particles of same energy gives mass resolution
- Proof-of-principle campaign at the University of Duisburg-Essen showed feasibility, however some improvements are still necessary:
  - Better knowledge of the ionisation volume position
  - Laser capable of ionising also Si, O

- Figure above: energy distribution of Mg sputtered by Ar ions from a MgSiO<sub>3</sub> thin film
- "Thomponic" behaviour (fit line) visible, even though high energetic particles not detected as efficiently

ToF-SNMS seems to be a well-suited experimental approach!

## Ejecta angular distributions

- Mass stuck to catcher QCM as a function of polar angle  $\beta$  (cf. sketch above)
- Reduction in sputtered mass for grazing impact and rough samples

QCM used to compare ejecta angular distributions across sample types

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